Photocatalytic oxidative desulfurization and degradation of organic
 pollutants under visible light using TiO₂ nanoparticles modified with iron
 and sulphate ions

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13 Abstract

The incorporation of iron and sulphate ions on TiO₂ nanoparticles has been carried out 14 15 in order to enhance catalytic performance in environmental remediation. The aim of this 16 work is to design a multifunctional and visible-light active photocatalyst to remove 17 pollutants from transportation fuels and water. The influence of doping with sulphur and iron on the physical, structural, optical, and electrochemical properties of TiO_2 has 18 19 been fully analysed by a wide variety of techniques. Thus, the incorporation of a low 20 amount of iron improves the photocatalytic performance, as in the case of 1.2%S-0.5%Fe-TiO₂ photocatalyst, which shows the highest activity under visible light 21 22 irradiation. The enhanced behaviour of the electron charge transfer of this photocatalyst 23 has been confirmed by electrochemistry measurements including linear sweep voltammetry and electrochemical impedance spectroscopy. Also, the stability and 24 recyclability of the best photocatalyst have been assessed. 25

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Keywords: Titanium dioxide, photocatalytic oxidative Desulfurization,
 photodegradation, dibenzothiophene, ciprofloxacin.

1 **1. Introduction**

2 The development of efficient and sustainable technologies for the reduction of pollutant gas emissions (CO_2 , NO_x , SO_x) and water pollution is one of today's greatest challenges. 3 The combustion of sulphur-containing compounds in fuel oils produces noxious sulphur 4 5 oxides (SO_x) which are directly released into the atmosphere. Thus, the reduction of sulphur content in fuels is urgently needed in order to avoid serious health and 6 7 environmental problems caused by toxic gas emissions [1] and also to comply with strict 8 regulations enforced for limiting sulphur content in liquid fuels [2]. In recent years, 9 conventional hydrodesulfurization (HDS) technology has been employed for sulphur removal in transportation fuels; however, it requires high hydrogen pressure and 10 11 elevated temperatures, which provoke an increase in the operating costs and loss of octane number [3]. Hence, photooxidative desulfurization has emerged as a promising 12 13 alternative to the hydrodesulfurization process. TiO₂-based materials have been proved to be efficient photocatalysts for the removal of different sulphur compounds from fuels 14 15 by desulfurization process [1]. Also, other systems based on composites have exhibited 16 excellent visible photocatalytic activity, including Ti₃C₂/g-C₃N₄ [4], ATP-CeO₂/MoS₂ [5],

17 CeO₂/ATP/g-C₃N₄ [6], MoO₃-MOF [7] and BiP_{1-x}V_xO₄/ATP [8].

On the other hand, emerging contaminants (including pharmaceutical, pesticides and 18 19 personal care products) have frequently been found in wastewater treatment plants, 20 even in freshwater, due to the fact that conventional treatments are ineffective for their 21 complete elimination [9]. Specifically, the presence of antibiotics in wastewater is producing undesirable human and environmental concerns such as antibiotic-resistant 22 23 bacteria and contamination of the aquatic environment [10]. In particular, fluorine-24 containing antibiotics (e.g. ciprofloxacin) are highly resistant to degradation and have been detected in higher concentrations than those of other drugs [11]. 25

To solve these challenges, efficient removal technologies are being developed to avoid the environmental effects of pollutants. Among all of these, photocatalysis has currently emerged as an effective, sustainable, and economic technology to reduce the environmental pollution. In this regard, titanium dioxide is one of the most promising and cost-effective photocatalysts for the environmental pollutants removal as it can be easily modified to enhance its photocatalytic properties under visible light irradiation.

1 One of the approaches that has proven to be promising in this enhancement is cation 2 and anion doping of TiO₂ due to the decrease in the band gap width and the suppression of electron-hole recombination [12]. To date, there are only a few studies that focus on 3 TiO₂ materials modified with iron and sulphur for photocatalytic experiments. For 4 5 example, Niu et al. [13] synthesized iron and sulphur co-doped TiO₂ photocatalysts by 6 solvothermal method with FeCl₃ and potassium persulfate as iron and sulphur sources, respectively. They found that TiO₂ material with a ratio of Fe/Ti = 1.0 % exhibited the 7 highest activity in phenol degradation. Co-doped S/Fe-TiO₂ materials synthesized by 8 9 Christoforidis et al. [14] achieved a high selectivity in the photodegradation of toluene which was attributed to the influence of sulphur on the properties of the TiO₂ surface. 10 These materials were prepared by microemulsion synthetic method using ammonium 11 sulphate and iron (III) nitrate as S and Fe sources, respectively. Microflow system was 12 13 employed by He et al. [15] to obtain active Fe-S-TiO₂ catalysts for the degradation of methyl orange using UV and visible light. More recently, Villaluz et al. have reported a 14 new approach to prepare Fe/N/S-TiO₂ materials using ammonium iron (II) sulphate for 15 16 removal of 4-chlorophenol [16]. To the best of our knowledge, there are no studies in the literature using iron and sulphate modified TiO₂ nanomaterials for both 17 18 photocatalytic removal of sulphur-containing pollutant and an antibiotic drug.

Herein, TiO_2 nanoparticles have been modified with iron, a highly abundant and sustainable metal, and sulphate ions by using $Fe(NO_3)_3$ and Na_2SO_4 as chemical sources, respectively. The as-synthesized materials have been completely characterized and tested as versatile photocatalysts in the elimination of dibenzothiophene, as well as, in the degradation of ciprofloxacin with visible light irradiation. The latter is a highly stable drug that has been studied by electrochemical methods under visible light in the presence of iron and sulphate modified TiO_2 .

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27 2. Experimental Section

28 2.1. Materials

Titanium (IV) isopropoxide (TIPT), nitric acid 65 %, ciprofloxacin (> 98 %), dibenzothiophene were purchased from Sigma Aldrich-Merck. Iron(III) nitrate—water (1/9) (Fe(NO₃)₃·9H₂O), sodium sulphate (Na₂SO₄) and hydrogen peroxide solution (H₂O₂) 30 % were purchased from Scharlab. Dodecane \geq 99.8 % was acquired from Alfa Aesar,

2-propanol from VWR Chemicals and n-octane from Fluorochem. Water was obtained
 from a Millipore Milli-Q system (Waters, USA).

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2.2. Synthesis of catalysts

2.2.1. Synthesis of TiO₂ nanoparticles modified with Fe³⁺ and SO₄²⁻ ions

5 The synthesis procedure of modified TiO₂ nanoparticles follows a typical sol-gel method, in which 40 mL of titanium (IV) isopropoxide were dissolved in 60 mL of 2-propanol and 6 7 added dropwise to an aqueous solution of HNO_3 at pH = 2. Fe(NO_3)₃·9H₂O and Na_2SO_4 were used as the source of Fe^{3+} and SO_4^{2-} , respectively, and added dropwise at the same 8 9 time to the acidic solution in different quantities (0.5-1 % for Fe and 0.6-1.5 % for S) 10 previously dissolved in 100 mL of milliQ water. The pale orange mixture was vigorously stirred for 16 hours. Then, the suspension was filtered, washed and dried. Finally, the 11 samples were calcined in air in an oven at 400 °C for 16 hours (0.8 °C/min) and labelled 12 13 as X%Fe-TiO₂, Y%S-TiO₂ and Y%S-X%Fe-TiO₂, where X and Y are the real percentages of Fe and S incorporated in each sample, respectively. 14

15 2.3. Characterization

16 All the synthesized nanoparticles were characterized using the following techniques: X-17 Ray diffraction (XRD) patterns were carried out with a Phillips Diffractometer model PW3040/00 X'Pert MPD/MRD at 45 KV, 40 mA with Cu-K α radiation (λ =1.5418 Å), 18 adsorption-desorption isotherms of nitrogen and pore size distributions were obtained 19 using a Micromeritics TriStar 3000 analyser applying the Barret- Joyner-Halenda (BJH) 20 21 model, FTIR spectra were recorded on a Varian Excalibur Series 3100 - UMA 600 spectrometer in Attenuated Total Reflectance (ATR) mode. DRUV-Vis spectra (in diffuse 22 23 reflectance mode) were obtained using a Varian Cary 500 spectrophotometer. The band 24 gap values were obtained applying the Kubelka-Munk function $[F(R_{\infty})]$ to DRUV-Vis 25 spectra. Thermogravimetric analysis was obtained under air atmosphere with a Star System Mettler Thermobalance in the range of 40 to 1000 ⁰C of temperature. 26 Transmission electron microscope (TEM) studies were performed using a 2100 JEM 27 (JEOL) electron microscope operating at 200 kV and Scanning Electron Microscope in 28 29 high resolution (SEM-FEG) using a Nova NanoSEM 230 microscope. Fe content was determined by induced coupled plasma atomic emission spectroscopy (ICP-AES) using a 30 31 7300DV optical emission spectrometer from Perkin Elmer and S content was calculated

1 by elemental analysis on a CHNOS model Vario EL III of Elemental Analyses System. The electrochemical measurements were recorded with a potentiostat/galvanostat Autolab 2 PGSTAT302 through modified carbon paste electrodes (MCPE) used as working 3 electrode with the same procedure than in previous works [17]. Electrochemical 4 impedance spectroscopy (EIS) data were recorded by using a potentiostat/galvanostat 5 6 with an impedance module with AC signal of 10 mV amplitude in the frequency range between 0.01 Hz and 100 kHz in potentiostatic conditions. The electrode potentials were 7 measured against a saturated Ag/AgCl(s) reference electrode. Some of the 8 9 electrochemical measurements were performed in phosphate buffer at pH 7.4 or 0.2 M Na₂SO₄ solution using deionized water. 10

11 **2.4.** Photocatalytic Desulfurization of dibenzothiophene.

12 In this experiment, 20 mg of photocatalyst, 10 mL of n-octane (model oil) containing dibenzothiophene (100 ppm S), 10 mL of acetonitrile and hydrogen peroxide 30 % in 13 14 different proportions were mixed in a 20 mL flask tube and stirred at room temperature under the irradiation of visible light produced via a 300 W Xenon lamp. After certain 15 16 time, the reaction mixture was centrifuged, and the supernatant was analyzed by GC-FID (Perkin-Elmer GC Clarus 580) with a Velocity[®] column (dimethylpolysiloxane, 30 m, 17 18 0.25 mm, 1.00 μ m) using dodecane as an external standard. The DBT conversion was calculated according to equation: Conversion (%) = [(initial mmoles of DBT - final mmoles 19 of DBT)/initial mmoles of DBT] × 100. 20

21 **2.5.** Photocatalytic degradation of ciprofloxacin (CIP).

To perform the degradation reaction, the photocatalyst (8 mg) was suspended in an aqueous CIP solution with a concentration of 15 ppm. Then, the mixture was stirred in the dark during 1 h with the aim to reach the adsorption equilibrium of CIP molecules in catalysts' surface. After that, the visible lamp ($\lambda > 420$ nm and 300 W) was switched on and an aliquot was repeatedly taken to obtain the corresponding kinetic curve. The degradation of CIP was followed using a spectrophotometer (SP-830) at 272 nm wavelength, which is the maximum absorbance of CIP molecule.

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30 3. Results and discussion

31 **3.1. Synthesis and characterization**

1 The modified TiO₂ nanoparticles have been synthesized through the hydrolysis of titanium isopropoxide as titanium source at controlled pH and using low amounts of Fe³⁺ 2 and SO₄²⁻ species. The aim of this strategy is to demonstrate that the simultaneous 3 addition of both ions exerts a synergetic influence over the photocatalytic properties in 4 the selected reactions. The quantities of modifiers have been selected in order to ensure 5 6 the suitable adjustment of the band gap value in the material so that it is active in the visible region of the electromagnetic spectrum and to avoid a high electron-hole 7 8 recombination rate. It is also desirable to attain the adequate textural properties. The 9 percentage of iron incorporation was established in a theoretical maximum of 1% based on some authors' studies such as Moradi et al., who demonstrated that a high 10 percentage of Fe is detrimental for the photocatalytic activity [18]. This group's studies 11 12 demonstrated that the degree of particle agglomeration increases with the doping of Fe into the TiO₂ lattice, which is not beneficial for the photocatalytic activity. Regarding the 13 14 incorporation of sulphur as sulphate ions, a maximum percentage of 1.5% S has been selected bearing in mind a previous report in which similar percentages of sulphur atoms 15 16 enhanced the selectivity towards the desired product in an oxidation reaction [19]. Other authors have studied the catalytic and photocatalytic activities with the increase 17 18 of sulphur doping in TiO₂ nanoparticles and have also concluded that a high S-content involves a non-significant enhancement of results or even a worsening of them [20]. 19 Boningari et al. observed a similar behaviour in the photocatalytic degradation of 20 21 acetaldehyde using materials with high sulphur loadings [21]. In conclusion, the 22 materials studied in this work are based on TiO₂ nanoparticles modified with SO_4^{2-} groups (0.6%S-TiO₂) or Fe³⁺ cations (0.4%Fe-TiO₂) and TiO₂ nanoparticles modified with 23 both Fe^{3+} and SO_4^{2-} ions (1.2%S-0.5%Fe-TiO₂ and 1.5%S-0.8%Fe-TiO₂). 24

Firstly, all the synthesized photocatalysts were fully characterized. The X-Ray diffraction patterns of 0.4%Fe-TiO₂, 0.6%S-TiO₂, 1.2%S-0.5%Fe-TiO₂ and 1.5%S-0.8%Fe-TiO₂ samples are shown in Fig. 1 and consist of anatase crystal form. The anatase (JCPDS pattern reference card No. 21-1272) [22] crystallizes in an octahedral system and the most intense peaks appear at 2θ values of 25° (101) and 48° (200). The temperature of calcination (400 °C) could mean the appearance of determined peaks of brookite phase as other authors affirm [15]. However, the incorporation of metal and/or non-metal

1 atoms into TiO₂ lattice, presumably, favours the formation of anatase phase [19]. Some 2 peaks corresponding with (105), (211), (116) and (220) planes are clearly defined in S or 3 Fe-modified TiO₂ samples while in S,Fe-TiO₂ samples the two peaks turn into a broader peak, due to the incorporation of two types of ions and the increase in the amount of 4 them. In addition, no other peak attributed to iron oxide can be detected in the XDR 5 patterns, so we could affirm that all Fe³⁺ ions have incorporated into TiO₂ lattice in 6 substitution of Ti⁴⁺ cations, due to their similar radii [18]. Zooming in on the most intense 7 8 peak area, a significant broadening and a slight shift towards higher diffraction angles of the 25° peak are observed the SO₄²⁻ ions are incorporated on TiO₂ in comparison to the 9 sample modified only with Fe that could be indicative of an external incorporation of 10 the SO₄²⁻ ions (see Fig. S1 and Table S1) [23]. According to Williamson and Hall, the 11 12 diffraction line broadening is due to crystallite size and strain contribution, which is provoked by the crystal imperfection and distortion in powders. The lattice strain can 13 be calculated by the following formula: 14

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$$\varepsilon = \frac{\beta}{4tan\theta}$$

where ε is the lattice strain, β is the full width at half maximum (FWHM) of the (101) diffraction peak and θ is the angle of diffraction. Thus, the lattice strain of 0.4%-Fe-TiO₂ was found to be 0.0178, a lower value in comparison to the values obtained for the materials modified with SO₄²⁻ species, which are superior to 0.02 (see Table S1). The broadening of the (101) peak and the increase in the lattice strain for the materials modified with SO₄²⁻ species imply a loss in the crystallinity and an increase in lattice disorder, due to the incorporation of the SO₄²⁻ ions on the TiO₂ surface.





Fig. 1. XRD patterns of the prepared TiO₂ samples.

3 Textural properties such as surface area, BJH average pore diameter and total pore 4 volume were obtained from nitrogen adsorption and desorption isotherms. As can be 5 seen from Fig. 2a, the samples exhibit similar distributions, well-assigned to TiO₂ 6 nanoparticles [19, 24] and attributed to the mesoporous structure. All materials show 7 type-IV isotherms with a H1 hysteresis loops according to IUPAC classification. However, 8 type-IV isotherms of 0.6% S-TiO₂ and 0.4% Fe-TiO₂ materials show high absorption 9 capability at high relative pressures (around 0.8), which indicates an increase in pore volume values, e.g., 0.6% S-TiO₂ sample with a pore volume of 0.26 cm³·g⁻¹ (Table 1). 10 The pore size distribution of the TiO₂ nanoparticles was calculated from the adsorption 11 branch of N₂ isotherm by the Barret, Joyner and Halenda (BJH) method (Fig. 2b) and 12 13 shows a narrow and homogeneous range for the single-modified samples around 9 nm. 14 With the introduction of both species, the pore size distribution decreases until values around 5 nm, which is related to higher superficial BET areas as can be seen in Table 1. 15





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Fig. 2. (a) N₂ adsorption-desorption isotherms and (b) pore size distribution of all prepared TiO₂ samples.

5 Representative TEM micrographs and the size distribution histograms of all modified TiO₂ nanoparticles are shown in Fig. 3 and Fig. S2. The typical rough morphology of TiO₂ 6 nanoparticles and the formation of mesoporous TiO₂ aggregates can be observed in TEM 7 images. SEM-EDS analysis (Fig. S3) confirms the presence of iron and sulphur in the 8 9 samples. The length and width ranged from 3-12 nm as the size distribution indicated. For the single-modified samples, a significant aggregation of the 0.4% Fe-TiO₂ 10 nanoparticles is observed with a particle size centred at 7.5 nm, while 0.6% S-TiO₂ 11 12 sample shows a lower aggregation and a smaller particle size at 6.7 nm. After the incorporation of both ions, 1.2%S-0.5%Fe-TiO₂ and 1.5%S-0.8%Fe-TiO₂ samples 13 maintained a narrow size distribution centred at 6.5 and 5.7 nm, respectively, although 14 the latter shows a slightly wider distribution related to its small size which could provoke 15 a certain aggregation of particles. Scherrer's equation was used to calculate the 16 crystallite size of nanoparticles from the greatest intensity XRD peak, and the results 17 were correlated to the particle sizes calculated from TEM images (Table 1). The SAED 18 19 patterns (inset in Fig. 3) indicate polycrystalline anatase nature of TiO₂ samples, which 20 is in good agreement with the XRD results.



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- 3

1.2%S-0.3%Fe-TiO₂, and d) 1.5%S-0.7%Fe-TiO₂ nanoparticles.

4 Table 1 shows a summary of the textural properties obtained by N₂ isotherms, contains the modifiers' quantities incorporated on titania network obtained by ICP analysis (Fe) 5 6 and elemental analysis (S) and also includes the particle size and band gap values for all nanoparticles. All synthesized TiO₂ nanoparticles exhibit large surface areas and pore 7 8 volume values due to the synthesis procedure and the significantly low temperature of calcination. A higher surface area is correlated with a lower particle size as can be 9 10 observed for 1.5%S-0.8%Fe-TiO₂ sample, while 0.4%Fe-TiO₂ sample possesses the lowest surface area and the highest particle size (9.5 nm). As can be seen, an increase of 11 S content provokes a gradual increase of the BET surface area values (from 128 m²/g to 12 144 m²/g). Christoforidis et al. pointed out that sulphur as dopant limits the formation 13 of anatase phase on TiO₂ nanoparticles, disfavours the formation of large TiO₂ 14 15 crystallites, and prevents the aggregates [14]. This agrees with TEM analysis and the 16 crystallite size results and justifies the slightly broadening of the most significant peak at 25 20 observed in XRD pattern (Fig. S1). Therefore, although the ionic radius of the 17 Fe³⁺ is close to Ti⁴⁺, a doping with Fe atoms into TiO₂ network also increases the final 18 superficial area BET in samples. Thus, a combination of both ions enhances the textural 19 properties of TiO₂ nanoparticles, which is one of the effective methods for increasing 20 21 photocatalytic activity.

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S Fe Particle Band V_p^a $\mathsf{D}_{\mathsf{p}}^{\mathsf{a}}$ S_{BET} Crystallite Material Content Content size gap $(m^2.g^{-1})$ $(cm^{3}.g^{-1})$ (nm) size (nm) (%)^b (eV)^d (%)^b (nm)^c 0.6%S-TiO₂ 128.1 0.26 8.6 -0.6 8.1 6.7±0.1 3.18 0.4%Fe-TiO₂ 107.2 0.24 9.1 0.43 9.5 7.9±0.2 2.94 6.5±0.2 7.6 1.2%S-0.5%Fe-TiO₂ 133.0 0.21 4.9 0.46 1.2 2.85 5.7±0.1 1.5%S-0.8%Fe-TiO₂ 144.4 0.19 4.7 0.81 1.5 6.9 2.73

2 Table 1. Textural properties of synthesized TiO₂ samples.

^a Total pore volume and pore size as calculated by the BJH method from the adsorption branch
 of the N₂ isotherm. ^b Fe and S contents were measured by induced coupled plasma atomic
 emission spectroscopy (ICP-OES) and elemental analysis, respectively. ^c Determined by
 Scherrer's equation. ^d Calculated by the application of the Kubelka-Munk equation.

The modified TiO₂ nanoparticles were also analyzed by UV-visible Diffuse Reflectance 9 10 (Fig. 4). All samples show a similar spectrum with a broad band around 320 nm associated with band-band transition, characteristic of these types of materials. The 11 absorbance of S,Fe-TiO₂ nanoparticles in the visible region increases and a red shift in 12 the absorption onset value is observed with increasing modifier percentage (Fig. 4a). 13 Kubelka-Munck equation was used to estimate the band gap energy of the synthesized 14 samples by plotting $(F(R) \cdot E)^{1/2}$ vs E (eV), assuming indirect band gap (see Fig. 4b). The 15 16 band gap energy values of all TiO₂ samples are summarized in Table 1, and as can be seen, the values decrease with increasing amount of modifier added, from 3.18 eV for 17 0.6%S-TiO₂ sample until 2.73 eV in the case of 1.5%S-0.8%Fe-TiO₂ sample. It is 18 remarkable that the incorporation of iron doping leads to a narrower band gap value 19 (2.94 eV) in comparison to the modification with sulphate ions (3.18 eV) in the single-20 modified samples. This may be explained by the incorporation of Fe⁺³ atoms into inner 21 TiO_2 network, which provokes a higher impact than the attachment of SO_4^{2-} groups on 22 23 the TiO₂ surface.





Fig. 4. (a) UV-Visible Diffuse reflectance spectra of TiO₂ samples and (b) plots of Kubelka-Munck
 equation transformation.

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The FTIR spectra of all synthesized TiO₂ nanoparticles were also recorded in the range 5 of 500-4000 cm⁻¹ (Fig. 5). All the samples show an intense and broad band around 3300 6 7 and at 1630 cm⁻¹ due to the surface adsorbed water and hydroxyl groups and correspond to the O-H bending vibration. The intense peak at 500 cm⁻¹ is associated to the stretching 8 vibration of Ti-O bond and the Fe-O bond. The bands at 983, 1046, 1124 and 1209 cm⁻¹ 9 (shoulder-type the latter peak) are characteristic of a bidentate SO₄²⁻ moiety 10 coordinated to Ti⁴⁺ of titania's surface. The two bands at 983 and 1046 cm⁻¹ are assigned 11 to symmetric and asymmetric stretching vibration of S-O bonds and the signals at 1124 12 and 1209 cm⁻¹ correspond to S=O vibrations [19, 20]. The intensity of these peaks is 13 greater in the sample with the highest sulphur content, 1.5%S-0.8%Fe-TiO₂. Thus, FTIR 14 15 analysis confirms that SO_4^{2-} ions are anchored on the surface of TiO₂.





Fig. 5. FTIR spectra of all TiO₂ samples.

3 Thermogravimetric analysis (TGA) of the S, Fe-TiO₂ samples was carried out with a heating rate of 5 °C/min from 40 °C to 1000 °C under N₂ gas atmosphere and the results 4 5 are shown in Fig. 6. The TGA-DTG thermograms show a total weight loss of 7 % and 8 % for 1.2%S-0.5%Fe-TiO₂ and 1.5%S-0.8%Fe-TiO₂ samples, respectively, and exhibit three 6 7 different regions of weight loss in each sample. The first region, located between 40 and 180 °C, is attributed to the removal of the physisorbed water on the nanoparticles' 8 9 surfaces. The second one is observed from 190 °C to 550 °C and is related to the decomposition of species, such as nitrates and sulphates, used in the synthesis of 10 nanoparticles [25]. Additionally, the intense peak in the DTG curves around 700 °C 11 corresponds to the phase transformation of TiO₂ from anatase to rutile [25]. 12



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Fig. 6. TGA plots of 1.2%S-0.5%Fe-TiO₂ (blue line) and 1.5%S-0.8%Fe-TiO₂ (green line),
and DTG curves of 1.2%S-0.5%Fe-TiO₂ (dotted black line) and 1.5%S-0.8%Fe-TiO₂ (solid
black line).

Fig. 7 shows the photoluminescence emission spectra of all synthesized TiO₂ 5 nanoparticles at the excitation wavelength of 250 nm. The spectra of all samples exhibit 6 7 several peaks where an intense band stands out around 400 nm associated to the band gap transition of TiO₂. In the wavelength range of 450–500 nm the presence of other 8 9 intense peaks is attributed to the defects/vacancies of oxygen on surface [17]. The PL 10 spectra provides additional information on the separation and recombination process 11 of the electron-hole pair and the efficiency of charge-carrier transfer. As can be observed, a significant drop in the fluorescence intensity of the S,Fe-modified TiO₂ 12 samples is recorded in comparison to the single-modified samples (0.6%S-TiO₂ and 13 0.4%Fe-TiO₂). This refers to the inhibition of the photo-generated electron 14 recombination that takes place from the conduction band to the valence band in TiO₂-15 based photocatalysts. Thus, the incorporation of iron and sulphate ions drives to a lower 16 17 recombination rate and more efficient separation of electron-hole pairs and reveals a 18 higher lifetime of photo-induced charge carriers [19].





Fig. 7. PL spectra of all synthesized TiO₂ samples.

3 In order to gain further insight into the chemical composition and chemical states of the synthesized TiO₂ nanoparticles, XPS analysis was performed on the S, Fe-modified 4 samples and the results are shown in Fig. 8. The wide-survey spectra of the analyzed 5 samples showed all the predominant peaks in the samples (Fig. 8a), even the elements 6 7 with low content. The carbon peak for C 1s located at 284.8 eV is due to the surface adventitious carbon and all electron binding energy were calibrated according to it. A 8 9 small peak corresponding to N 1s around 400 eV confirms the presence of nitrogen 10 moleties that could be assigned to chemisorbed nitrogen such as N_2 species [26].

11 The spectra of Ti 2p, S 2p and Fe 2p are separately depicted in Fig. 8 b, c, d, respectively. 12 Fig. 8b shows the two components of Ti 2p, $2p_{3/2}$ and $2p_{1/2}$, at 458.8 eV and 464.6 eV of binding energy, respectively. These peaks were assigned to the oxidation state of 13 titanium Ti⁴⁺ and correlated with the formation of stoichiometric TiO₂ material in a 14 tetragonal structure since the photopeaks revealed a high symmetry and a lack of 15 shoulders on the lower energy sides [27]. For the S 2p spectra signals were observed 16 17 around 169 eV (doublet $2p_{3/2}$ at 168.4 eV and $2p_{1/2}$ at 169.6 eV) with the S atom in the 18 (VI) oxidation state corresponding to sulphate groups anchored to the TiO₂ surface 19 coordinating Ti⁴⁺ species [28].

1 The XPS signals of Fe 2p are weaker and less clear than the peaks of S 2p or Ti 2p, (Fig. 2 8d). The main characteristic peaks of Fe 2p, $2p_{3/2}$ and $2p_{1/2}$, appear at 710.1 and 723.7 eV, with the presence of satellite peaks around 716.4 eV, which is consistent with the 3 existence of iron in the positive trivalent form, Fe³⁺. By comparing the XPS peaks and the 4 obtained oxidation states of Ti⁴⁺ and Fe³⁺, we can conclude that the replacement of Ti⁴⁺ 5 ions from Fe³⁺ occurs and the Fe³⁺ ions can be incorporated into the TiO₂ lattice to form 6 Ti-O-Fe bonds (similar radii, 0.65 Å for Fe³⁺ and 0.606 Å for Ti⁴⁺ and electronegativity) 7 8 [29].



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Fig. 8. XPS spectra of the S,Fe-modified TiO₂ samples.

The chemical-physical characterization of the synthesized samples was also studied. The electrochemical properties of our photocatalysts in phosphate buffered saline at pH 7.4 were examined. As an example, 1.5%S-0.8%Fe-TiO₂ material was used to prepare a modified carbon paste electrode (MCPE), immersed in PBS and measured by cyclic voltammetry (CV). Fig. 9 shows cyclic voltammograms for working electrode modified with TiO₂ sample using platinum as a counter and Ag/AgCl(s)/KCl (3M) as a reference

1 electrode. As can be observed, typical voltammograms for bulk TiO₂ nanoparticles were 2 recorded with a pronounced anodic peak by scanning in the positive direction associated 3 to the electrochemical formation of Ti(III) species in TiO₂ nanoparticles' surfaces. Scanning towards negative potentials it can be inferred that the cathodic peak overlaps 4 5 with the reduction limit imposed by the solvent. The reactions that take place are: $Ti(IV)(TiO_2) + e^- + H^+ \text{ or } M^+(aq) \rightarrow Ti(III)(TiO_2) + H^+ \text{ or } M^+(TiO_2)$. By increasing the scan rate, 6 the current intensity of voltammograms increases, but the shape is maintained, which 7 reveals a semi diffusion-controlled process by the electrolyte cation insertion into the 8 9 outer network. Due to this, the TiO₂ modified electrode can be considered as a quasi-10 reversible system with the presence of reversible surface states.



Fig. 9. CV obtained for MCPE with 1.5%S-0.8%Fe-TiO₂ sample immersed in aqueous 0.1 M
 phosphate buffer with a reversal potential V = -1.5 V at different scan rates vs Ag/AgCl(s)/KCl
 (3M) as reference electrode.

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With the aim to prove the presence of Fe³⁺ cations into TiO₂ network, differential pulse voltammetry (DPV) technique was performed between -0.4 V and 0.5 V and the results shown in Fig. 10. This electrochemical technique is much more sensitive than cyclic voltammetry and allows us to obtain more information of the modified TiO₂ nanoparticles. In this case, 1.2%S-0.5%Fe-TiO₂ and 1.5%S-0.8%Fe-TiO₂ samples were recorded. The cathodic current (from positive to negative potential values) reveals a reduction peak around 0.09 V consistent with the reduction of Fe³⁺ to Fe²⁺ species. After returning the potential, an anodic peak is clearly observed at potential values around 0.06 V corresponding to the reversible oxidation to Fe³⁺ cations. This behaviour is identical for both S,Fe-modified samples, observing a slight shift and a higher current intensity for the sample with the highest percentage of iron, 1.5%S-0.8%Fe-TiO₂, which supports that the iron loading does not render different structural forms on the titania network.



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Fig. 10. Anodic and cathodic differential pulse voltammograms (75 mV modulation amplitude)
 of both S,Fe-modified TiO₂ samples (10 % to graphite) after immersion in phosphate buffer as
 electrolyte and Ag/AgCl/KCl (3M) as reference electrode.

Electrochemical characterization of the modified TiO₂ nanoparticles was completed by means of cyclic voltammetry measurements using modified carbon paste electrodes (MCPE) as were prepared previously [17, 24]. Thus, the HOMO/LUMO energy levels positions of 1.2%S-0.5%Fe-TiO₂ and 1.5%S-0.8%Fe-TiO₂ materials were calculated based on the following equations: $E_{LUMO} = -(E_{red} + 4.4) eV$

18
$$E_{HOMO} = -(E_{ox} + 4.4) eV$$

E_{red} and E_{ox} are onset reduction and oxidation potentials, and 4.4 is the energy level of the reference electrode used as standard, Ag/AgCl/KCl (3M). The optical band gap of the samples (E_g) can be calculated through E_g = $1240/\lambda_g$, where λ_g is the absorption edge of the material calculated by DRUV spectra (Fig. 3). With all these data, E_{HOMO} and E_{LUMO} are estimated to be -6.70 and -3.68 eV, and -6.64 and -3.71 eV, for 1.2%S-0.5%Fe-TiO₂ and 1.5%S-0.8%Fe-TiO₂, respectively.

4 3.2. Oxidative desulfurization of dibenzothiophene (DBT).

5 The catalytic activity of 1.2%S-0.5%Fe-TiO₂ nanoparticles was investigated for DBT 6 removal in 10 mL of model oil containing 100 ppmS at the selected conditions: time 3h, 7 room temperature, 20 mg of catalyst, and in presence of 10 mL of acetonitrile. The 8 experiment results are shown in Table 2. First of all, the result of Entry 1 demonstrated 9 the effect of the addition of acetonitrile to the mixture reaction as multiple studies have 10 pointed out [30, 31]. Dibenzothiophene molecules were extracted from the model oil 11 (n-octane) to the polar phase (acetonitrile) in a percentage of 55.1 %. When hydrogen peroxide was added, only 64.4 % of conversion was reached due to the oxidation of DBT 12 molecules in 3 hours without catalyst. The next experiment was carried out in presence 13 of 1.2%S-0.5%Fe-TiO₂ photocatalyst without visible light irradiation (Entry 3) showing no 14 noteworthy increase in DBT conversion (74.0 %). However, when the visible lamp was 15 16 turned on, the conversion of DBT was accomplished by above 82 %, in a photooxidation 17 process. The influence of the different catalysts used was also investigated. As shown in 18 Fig. S4, the single-modified samples (0.6%S-TiO₂ and 0.4%Fe-TiO₂) perform a similar pattern in photocatalytic DBT oxidation reaching conversion percentages of up to 70 %. 19 However, the S,Fe-modified samples show greater DBT conversion values than those 20 obtained for single-modified samples up to 82.7 % with 1.2%S-0.5%Fe-TiO₂ sample. 21 22 These results may be due to the synergetic combination of both ions, a higher amount of sulphate ions in samples, which is beneficial in oxidation of sulphides [19], and a 23 24 limited percentage of iron in them. In fact, as other studies have confirmed, the 25 photocatalyst with the highest content of iron does not give the best result in DBT oxidation [32]. As Kalantari et al. affirmed, low concentrations of Fe³⁺ trap the electron 26 promoting the reduction of molecular oxygen and, with that, inhibit the recombination 27 electron-hole pairs and increase the photocatalytic activity. Nonetheless, a higher iron 28 concentration decreases the distance between trapping sites and Fe³⁺ cations can act as 29 a recombination centre and reduce the photocatalytic effect. 30

Table 2. Sulphur removal efficiency with different reaction conditions.

Entry	Catalyst	Visible light irradiation	O/S molar ratio	Conversion (%) ^a	Selectivity (%) ^b
1	-	No	-	55.1	100
2	-	No	4	64.4	100
3	1.2%S-0.5%Fe-TiO ₂	No	4	74.0	100
4		Yes	4	82.7	100
5		Yes	6	82.6	100
6		Yes	10	82.5	100

Experimental conditions: 10 mL model oil, 10 mL MeCN, room temperature, 20 mg of 1.2%S-0.5%Fe-TiO₂
 catalyst, time = 3 h. ^a Conversion and selectivity determined by GC. ^b Sulphone selectivity checked by ¹H
 NMR (Fig. S5).

5 The effect of time of light exposition and the catalyst dosage were performed using 6 1.2%S-0.5%Fe-TiO₂ sample in 20 mL of model mixture (n-octane + acetonitrile) and O/S 7 = 4 molar ratio, and the results are shown in Fig. S6. The catalytic results show that the DBT removal efficiency increases with the reaction time (see Fig. S6a), reaching the 8 maximum of conversion at 3 hours of light exposition. A slight decrease is observed with 9 10 a prolonged reaction time, which indicates a negligible effect of the time on the DBT desulfurization reaction after 3 h. Fig. S6b shows the effect of the catalyst mass in the 11 12 photooxidation of DBT. For the indicated material, the conversion increases with its 13 concentration within a limited range from 10 to 20 mg. However, the increase of the catalyst dosage above 20 mg does not mean an increase in DBT conversion. Some 14 authors have found a slight decrease in the activity at higher catalyst masses, which is 15 attributed to a poor dispersion of the catalyst or to the formation of an opaque 16 suspension in the reactor vessel, hindering the adsorption and diffusion of DBT and 17 18 inhibiting the excitation of the catalyst [33-35].

The addition of an oxidant agent as H_2O_2 has proven to enhance the photooxidation efficiency (see Table 2). With a molar ratio of O/S = 4 the sulphur conversion was 82.7 % but no further improvement was observed when the O/S molar ratio was increased to O/S = 6 or O/S = 10 ratios (Entries 5 and 6, respectively). Several reasons can explain this

1 behaviour, the oxidation process can be hampered by the presence of water in high quantity which would decrease the emulsion properties of the fluid [30, 36], or by the 2 3 addition of H₂O₂ which may form a liquid film on the catalyst surface hindering DBT adsorption to the active sites [37] and deactivating the material surface [38]. Other 4 authors pointed out that the hydroxyl radicals generated when H₂O₂ is exposed to light 5 6 can be consumed by H_2O_2 molecules themselves as described in the following reaction: $H_2O_2 + OH \rightarrow H_2O + O_2^- + H^+$ [30]. For all these reasons, we tested the use of a promoter 7 8 or co-oxidant to enhance the DBT conversion. In the presence of hydrogen peroxide, peroxyacetic acid would be produced, which would coordinate to the Fe or/and Ti atoms 9 in the materials to form the so-called peroxometallic complexes and generate reactive 10 oxygen species on the catalyst surface [33]. As can be seen in Fig. 11 the effect of adding 11 12 acetic acid to the mixture reaction is beneficial since an increase of acetic acid means an increase of DBT conversion. The best result was 91.3 % of conversion which was reached 13 with a O/S ratio of 10 and 20 μ L of acetic acid. Comparing with other reported results, 14 Kalantari et al. obtained 40.3 % of DBT conversion in presence of N doped TiO_2 15 16 nanoparticles [39] and 59.8 % in the case of co-doped Fe-N-TiO₂ nanoparticles [32]. More recently, Guo et al. have demonstrated that dye-sensitized TiO₂@SBA-15 17 18 composites showed a higher photocatalytic activity than the unsensitised ones, obtaining 96.1 % of DBT conversion [34]. 19



Fig. 11. Effect of the use of H₂O₂ and acetic acid as oxidant agents. Conditions: 20 mg of 1.2%S 0.5%Fe-TiO₂ catalyst, 10 mL n-octane, 10 mL acetonitrile, room temperature, 3 h of visible
 lamp.

4

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3.3. Photocatalytic degradation of ciprofloxacin under visible light

Degradation of ciprofloxacin drug was evaluated in the presence of the TiO₂ 6 photocatalysts under visible light irradiation using 8 mg of photocatalyst and 20 mL of 7 8 CIP solution with 15 ppm of concentration and the results were depicted in Fig. 12. 9 Previously, the catalyst was mixed and stirred with the aqueous solution of ciprofloxacin 10 for 60 minutes to ensure the adsorption equilibrium on the photocatalysts' surface. Fig. 12a shows the quantity of drug adsorbed, which is similar to samples doped with iron 11 12 (around 21.5 %) and slightly lower than S-modified TiO_2 (14 %). Two additional experiments with the S,Fe samples were performed using 20 mg of photocatalyst and 13 14 the same CIP solution volume resulting a concentration of 1 g/L and provoking an increase in drug quantity adsorbed (around 36 %). Additionally, as we demonstrated in 15 16 previous reports, the photolysis of ciprofloxacin molecules was practically negligible 17 under similar experimental conditions in absence of the catalyst [19].





Fig. 12. Degradation kinetics of ciprofloxacin in terms of (a) concentration and (b) ln(C/C_o)
 versus time in minutes.

As can be seen in Fig. 12a, both S,Fe-TiO₂ photocatalysts have a higher degradation activity than that achieved for 0.6% S-TiO₂ or 0.4% Fe-TiO₂ samples under visible light irradiation. Thus, 0.6%-TiO₂ and 0.4%-TiO₂ nanoparticles achieved around 70 % of degradation at 2 hours of starting the experiment, while S,Fe-modified samples reached an elimination above 84 % at the same time, indicating a synergetic effect of the

1 incorporation of sulphate and iron ions to TiO₂. With the aim of ensuring the complete 2 degradation of CIP molecules, the ratio photocatalyst/volume was increased to 1 g/L for the S,Fe-TiO₂ samples. The best result was obtained for the 1.2%S-0.5%Fe-TiO₂ 3 photocatalyst, which was capable of degrading almost completely the 100 % of CIP 4 pollutant in 90 minutes. According to the data plotted in Fig. 12b, the photocatalytic 5 degradation of ciprofloxacin drug followed a pseudo-first-order kinetic for all the 6 prepared TiO₂ samples. The kinetic constants obtained in each experiment are given in 7 8 Table 3, where the promising results for S,Fe-TiO₂ samples are corroborated, reaching a rate of 0.034 min⁻¹ for 1.2%S-0.5%Fe-TiO₂ photocatalyst, which is nearly 4 times higher 9 10 than the results obtained for single-modified samples. The results obtained by other comparative systems reported are shown in Table 3. 11

 Table 3. Kinetic constants of pseudo-first order reaction of different catalysts for

 photocatalytic degradation of CIP.

Catalyst	<i>k</i> (min ⁻¹)	Catalyst dosage (g/L)	CIP concentration (ppm)	Reference
0.4%Fe-TiO ₂	0.009	0.4	15	This work
0.6%S-TiO ₂	0.009	0.4	15	This work
1.2%S-0.5%Fe-TiO ₂	0.016	0.4	15	This work
1.5%S-0.8%Fe-TiO ₂	0.015	0.4	15	This work
1.2%S-0.5%Fe-TiO ₂	0.034	1	15	This work
1.5%S-0.8%Fe-TiO ₂	0.022	1	15	This work
1%Ce-TiO ₂	0.0266	0.5	10	[40]
1%B-TiO ₂	0.0249	1	10	[40]
NaCl/TiO ₂	0.0099	1	10	[41]
N,S-TiO ₂	0.0115	0.5	10	[42]
N-TiO ₂	0.002	0.5	15	[43]
2.5%N-0.5%Fe-TiO ₂	0.0038	1	20	[26]
2.5%N-1%Fe-TiO ₂	0.0044	1	20	[26]
Ti ³⁺ /N-TiO ₂	0.0132	0.43	4.5	[44]

1 With the aim of justifying the different photocatalytic behaviour between the two S,Fe-2 modified TiO₂ samples in the DBT oxidation and CIP degradation, electrochemistry 3 measurements have been employed. Linear sweep voltammetry (LSV) was the selected technique to measure the photoelectrochemical behaviour of both S,Fe-TiO₂ catalysts 4 modified with carbon paste electrodes (MCPE). Fig. 13 shows the voltammograms 5 6 obtained in presence and in absence of visible light and by using Na₂SO₄ 0.2 M as an electrolyte. The increase observed in the photocurrent towards more positive potential 7 8 values is related to the higher and effective charge density and the lower electron-hole recombination. Iron doping seems to act as an electron-hole trap, and it inhibits the 9 recombination rate of the photogenerated electrons. Besides, the replacement of Ti⁺⁴ 10 by Fe⁺³ ions leads to Fe⁺³ states that are located near to the conduction level [32], which 11 results in the enhancement in the photocurrent [45]. Fig. 13 shows a better performance 12 of 1.2%S-0.5%Fe-TiO₂ in terms of enhancement of photocurrent than 1.5%S-0.8%Fe-13 TiO₂ photocatalyst both in darkness and under visible light irradiation. These results 14 validate the better catalytic and photocatalytic behaviour of 1.2%S-0.5%Fe-TiO₂ sample 15 16 in comparison to 1.5%S-0.8%Fe-TiO₂, despite the lower band gap value that the latter 17 possesses.



Fig. 13. Linear sweep voltammetry of both S,Fe-modified TiO₂ samples after immersion in
 Na₂SO₄ 0.2 M and Ag/AgCl/KCl (3M) as reference electrode.

1 To evaluate the resistance against electron transfer, cyclic voltammetry and electrochemical impedance spectroscopy (EIS) measurements of the samples modified 2 3 with sulphate and iron were also conducted and the results are shown in Fig. 14. The measurements were performed by using the modified sample with carbon paste 4 electrodes (MCPE) immersed in Na₂SO₄ 0.2 M and by using $[Fe(CN)_6]^{3-}$ 5 mM as a probe 5 6 molecule. Cyclic voltammograms (Fig. 14a) show how the modification of the carbon electrode resulted in a higher electron transfer mobility, i.e., S,Fe-modified TiO_2 7 8 nanoparticles electrodes improve the electron transfer since their cyclic voltammograms show a lower peak potential separation (ΔE) and a narrower intensity 9 peak. Regarding the EIS results (Fig. 14b), in the Nyquist plot, the semicircles at high 10 frequencies and medium frequencies are attributed to the charge-transfer resistance on 11 12 the electrode-electrolyte interface and the straight sloping designated to Warburg impedance is attributed to diffusional impedance. Compared to bare carbon paste 13 electrode (CPE), S,Fe-modified TiO₂ nanoparticles show a smaller internal resistance, 14 which means a better charge transport. Among them, 1.2%S-0.5%Fe-TiO₂ catalyst 15 16 presents the best performance in terms of charge transfer and improved photoactivity. According to the modified Randles equivalent circuit (inset in Fig. 14b), the electron 17 18 transfer resistance (R_{et}) for bare CPE is about 11.5 k Ω , for 1.5%S-0.8%Fe-TiO₂ catalyst is about 10.2 k Ω and for 1.2%S-0.5%Fe-TiO₂ catalyst is about 9.7 k Ω , revealing a better 19 electronic and ionic conduction in the latter, which demonstrates the best 20 electrochemical behaviour and photocatalytic activity of this catalyst. 21



Fig. 14. (a) CV and (b) EIS measurements of bare and modified carbon paste electrodes after
 immersion in Na₂SO₄ 0.2 M with Ag/AgCl/KCl (3M) as reference electrode and [Fe(CN)₆]³⁻
 5 mM as a probe molecule.

3

4

3.4. Mechanism proposal

In agreement with the results obtained, a probable mechanism for the photocatalytic 5 removal of DBT and CIP pollutants is proposed in Scheme 1. The process starts with the 6 light irradiation to the photocatalyst that, after visible absorption, promotes the 7 8 electrons from valence band to conduction band generating positives holes in the first 9 one. Electrons in conduction band are used for reducing molecular oxygen to give 10 superoxide radicals. It has been demonstrated that dibenzothiophene molecules can be oxidized to dibenzothiophene sulphone by both photogenerated positive holes and 11 superoxide radicals [46]. Similarly, ciprofloxacin molecule has been proven to be 12 degraded by these species, but hydroxyl radicals also play an important role in the 13 removal of CIP [44]. In the case of DBT removal, hydrogen peroxide can be reduced to 14 15 form hydroxyl radicals which play an important role in DBT photooxidation as reactive oxidizing agents like other studies have confirmed [47]. 16





Scheme 1. Schematic diagram for the photocatalytic removal of organic pollutants.

Additionally, to investigate the nature of the species generated during photocatalytic
experiments, a degradation kinetic of ciprofloxacin was performed and several aliquots
were taken at certain times and analyzed by cyclic voltammetry. As shown in Fig. S7,
before the photodegradation process starts, a clear oxidation peak appears around -0.7
V which can be associated to ciprofloxacin molecules. The intensity of this peak

decreases over time when the degradation takes place while, at the same time, the
appearance of new species with less negative reduction peaks is observed, that can be
justified by the formation of more oxidized species after the photooxidation process to
which the starting species have been subjected.

5 For a deeper understanding of the photocatalytic degradation pathways of ciprofloxacin by of 1.2%S-0.5%Fe-TiO₂ material the generated intermediate products during the 6 7 photocatalytic process under the optimum conditions were examined by HPLC/MS. Fig. 8 S8 shows the peaks obtained by HPLC analysis and the molecular weight of intermediate 9 products, respectively. Based on the experimental results and the literature, the possible degradation pathway of CIP molecules was proposed as following. In the first 10 place, defluorination process results in the replacement of F^- with OH⁻ (m/z 328) [48], 11 which evolves into m/z of 301 through decarboxylation in quinoline moieties. The next 12 pathway could go forward through two possibilities: via the elimination of one of the 13 hydroxyl groups (m/z 285) or through opening piperazine ring (m/z 275) [40]. The first 14 15 one could lose simultaneously another hydroxyl and carbonyl groups and open 16 piperazine ring to form m/z of 229 and the latter compound appears with the 17 unsaturation of two bonds in piperazine moiety and hydroxyl loss (m/z 255). Both compounds *m*/z 255 and 229 evolve into the same quinoline ring with *m*/z of 131, which 18 19 leads into m/z of 121 through an opening of the right ring. Finally, it leads to compounds with lower molecular weight until the complete mineralization into $CO_2 + H_2O$ (Fig. 15). 20



- 1
- 2

Fig. 15. Proposal for the CIP photodegradation mechanism.

3 3.5. Photocatalytic stability and catalyst' recycling.

Recyclability is a crucial property of the photocatalyst, which directly affects the practical
application of it. To investigate the stability of the best catalyst, 1.2%S-0.5%Fe-TiO₂
sample, the DBT and CIP photocatalytic experiments were repeated for several cycles.
After each photodegradation reaction, the selected photocatalyst was recycled through
filtration, washing with water and ethanol, and drying under vacuum for the next cycle.
As shown in Fig. S9, the activity of 1.2%S-0.5%Fe-TiO₂ was preserved after various cycles
for both photocatalytic reactions, revealing the good stability and recyclability of our

1 modified TiO₂ nanoparticles. The reused 1.2%S-0.5%Fe-TiO₂ catalyst was analyzed by 2 FTIR and differential pulse voltammetry (DPV) studies and the results are shown in Fig. S9 c, d, respectively. FTIR spectrum of the recovered materials shows the presence of 3 the bands associated to sulphate groups and DPV curves confirm the presence of Fe³⁺ 4 ions in titania network. This highly sensitive technique allows us to detect a slight shift 5 6 to more negative potential values, which could be assigned to the peroxometallic complexes formed on the catalyst surface due to the presence of oxidant species in the 7 reaction medium and that we inferred in the DBT oxidation discussion [49]. 8

9 4. Conclusions.

Photocatalysts based on the modification of TiO₂ nanoparticles with Fe³⁺ and SO₄²⁻ ions 10 have been prepared and have proved their efficiency for the photocatalytic removal of 11 two types of organic pollutants under visible light irradiation. The suitable adjustment 12 13 of the amounts of doping agents has allowed for designing multifunctional photocatalysts which are active in the visible region of the spectrum and show an 14 enhanced photocatalytic activity. Among them, 1.2%S-0.5%Fe-TiO₂ photocatalyst stands 15 16 out by a lower recombination rate and more efficient separation of electron-hole pairs 17 and reveals a higher lifetime of photo-induced charge carriers, which are corroborated by optical and electrochemical techniques, such as photoluminescence and 18 electrochemical impedance spectroscopy (EIS). 1.2%S-0.5%Fe-TiO₂ photocatalyst 19 accomplishes a 91.3 % of DBT conversion under optimized conditions, degrades CIP 20 molecules 4 times faster than single-modified TiO₂ nanoparticles, and maintains its 21 22 stability under several cycles. Cyclic voltammetry and HPLC-MS measurements confirm the formation of oxidized species after the photooxidation process to which the starting 23 24 species were subjected.

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